

Potassium ion conductivity of KNO_2 mixed oxides

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Received 30 July 2004; received in revised form 1 December 2004; accepted 15 December 2004
Available online 8 June 2005

Abstract

Two types of polycrystalline solids, KNO_2 mixed Gd_2O_3 and KNO_2 mixed Nb_2O_5 , were prepared. In the case for KNO_2 mixed Gd_2O_3 , a remarkable enhancement in K^+ ion conductivity was successfully realized by interstitially dissolved KNO_3 formed during the preparation process in the C-type cubic Gd_2O_3 crystal lattice. In contrast, the conductivity of KNO_2 mixed Nb_2O_5 was more than three orders of magnitude below that of the Gd_2O_3 – KNO_3 solid solution due to the KNO_3 decomposition during the preparation procedure. The K^+ ion conductivity of the polycrystalline Gd_2O_3 – KNO_3 solid solution is more than three orders of magnitude higher than that of the K^+ ion conducting K_2SO_4 polycrystal and the value exceeds that of K^+ - β'' -alumina single crystal in *ab* plane, realizing the highest ion conductivity among the K^+ ion conducting solid electrolyte series. The superior ion conducting properties are realized by forming the interstitially KNO_3 dissolved Gd_2O_3 solid solution, without decomposing the mixed salt with K^+ .

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Keywords: Solid electrolyte; Potassium ion; C-type rare earth oxide; Gadolinium oxide; Niobium oxide

1. Introduction

K^+ ion is one of conducting ion species in solids [1,2]. However, the K^+ ion (ionic radius (four-coordinate): 0.137 nm) conductivity is appreciably below the practical application region compared with those of Li^+ (ionic radius (four-coordinate): 0.059 nm) and Na^+ (ionic radius (four-coordinate): 0.099 nm) ion conductors, such as Li^+ ion super ionic conductor of LISICON [3] and Na^+ ion super ionic conductor of NASICON [4], according to its relatively large ionic size. One exception is the single crystal of K^+ - β'' -alumina [5] and its K^+ ion conductivity (in the *ab* plane) is as high as $10^{-1} \text{ S cm}^{-1}$ at 600°C and the value is comparable to those of well-known Li^+ or Na^+ ion conducting solids. However, the K^+ ion conductivity of K^+ - β'' -alumina single crystal along *c*-axis is several orders of magnitude lower than that in *ab* plane, indicating that the K^+ ion conductivity of the polycrystalline K^+ - β'' -alumina is

considerably below that of the K^+ - β'' -alumina single crystal. One of the other K^+ ion conductors is polycrystalline potassium sulfate (K_2SO_4). However, the K^+ ion conductivity of K_2SO_4 polycrystal is as small as $3.38 \times 10^{-5} \text{ S cm}^{-1}$ at 600°C [2] and still three orders of magnitude lower than that of the practical application region ($>10^{-2} \text{ S cm}^{-1}$) of the representative Li^+ and Na^+ ion conducting solids. In order to obtain practical K^+ ion conductors, the realization of polycrystalline K^+ ion conducting solid showing the ion conductivity which enters into the application region, is highly requested. In our previous communication [6], C-type Gd_2O_3 was selected as the mother polycrystalline solid which possesses the highest symmetric cubic phase and holds reasonable space for bulky ion conduction like K^+ ions.

In this study, two types of oxides, cubic Gd_2O_3 (C-type rare earth oxide) and monoclinic Nb_2O_5 , were selected as the mother phase candidates and the mixing of KNO_2 was carried out to clarify the formation of the oxide– KNO_3 solid solution and their K^+ ion conducting characteristics were compared in detail.

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2. Experimental

Gd₂O₃ and KNO₂ powders were mixed by ball milling method (FRITSCH GmbH, Pulverisette 7). The mixture was pelletized and heated at 450 °C for 12 h in air. The pellets were pulverized and the resulting powder was pelletized again, and heated at 600 °C for 12 h in air, and then sintered at 600 °C for 12 h in air. The preparation of the KNO₂ mixed Nb₂O₅ solids was also carried out in a similar manner.

The sample characterization was conducted by X-ray powder diffraction using Cu K α radiation (Rigaku, Multiflex). The XRD data were collected by a step-scanning method in the 2θ range of 10–70°. The composition was identified by X-ray fluorescence analysis (Rigaku, ZSX100e). Fourier transform-IR spectrometer (JASCO International Co. Ltd., FT/IR-430) was used to obtain the infrared spectra of the samples. The electrical ac conductivity was measured using the sample pellet with the two Au electrodes by the complex impedance method (Hewlett-Packard, impedance analyzer 4192A) at the frequency region 5 Hz–13 MHz in the temperature range between 200 and 500 °C.

3. Results and discussion

Figs. 1(a) and 2 present the X-ray powder diffraction patterns of the KNO₂ mixed Gd₂O₃ and the KNO₂ mixed Nb₂O₅ series, respectively. Up to $x=0.405$ in $(1-x)\text{Gd}_2\text{O}_3-x\text{KNO}_3$, only C-type cubic Gd₂O₃ phase was observed, while additional peak corresponds to KNO₃ was observed for $x>0.405$. Furthermore, the lattice parameter of C-type cubic Gd₂O₃ monotonously increases with the KNO₃ content in $(1-x)\text{Gd}_2\text{O}_3-x\text{KNO}_3$ as shown in Fig. 1(b), suggesting the formation of the Gd₂O₃–KNO₃ solid solution. From Fig. 1, it becomes clear that the solid solution limit in the series is estimated to be ca. $x=0.405$. In contrast, for the KNO₂ mixed Nb₂O₅ series, all samples prepared show peaks corresponding only to mother Nb₂O₅ phase without any peak shift in the XRD patterns.

From IR measurements of $(1-x)\text{Gd}_2\text{O}_3-x\text{KNO}_3$ with $x=0.405$ and 0.469 (Fig. 3), the bands at 1384 cm⁻¹ (ν_3 asymmetric stretching mode of NO₃⁻) and 827 cm⁻¹ (out-of-plane ν_2 deformation mode of NO₃⁻) were observed for both samples, suggesting that NO₃⁻ anion exists in both solids. Furthermore, we have confirmed that the IR spectra of mother Gd₂O₃ did not change by KNO₂ doping. Therefore, it is considered that KNO₂ may be oxidized by ambient oxygen during the heating process. Since it is evident from Fig. 1 that the solids for $x \leq 0.405$ are the single solid solution, while the solid is the mixture phase of the Gd₂O₃–KNO₃ solid solution and KNO₃ phases for $x \geq 0.469$, the IR spectrum for the solid solution of $x=0.405$ is the NO₃⁻ anion existing in the interstitial sites of the solid solution. From the IR measurements for the KNO₂ mixed Nb₂O₅ series, there appears no peak corresponding to NO₃⁻ anion, indicating that all the samples prepared in the KNO₂ mixed Nb₂O₅ series do not

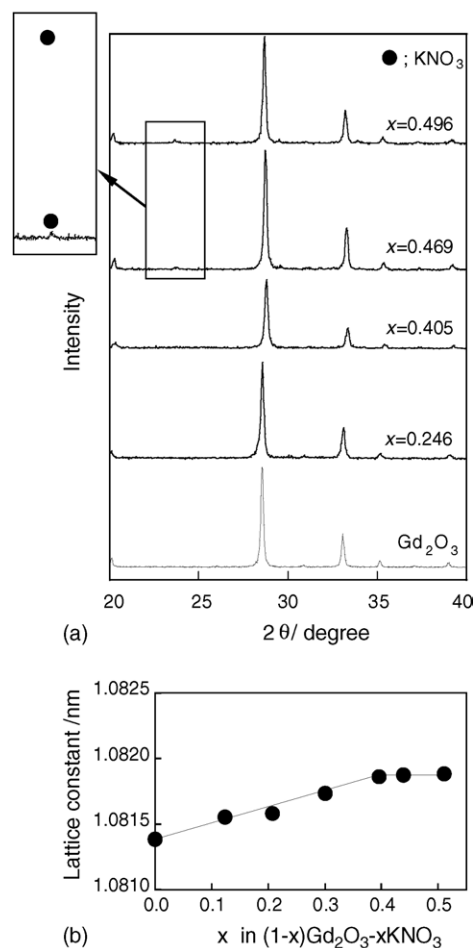


Fig. 1. (a) The X-ray powder diffraction patterns of KNO₂ mixed Gd₂O₃ and (b) the relationship between lattice parameter and the KNO₃ content in the Gd₂O₃–KNO₃ solids.

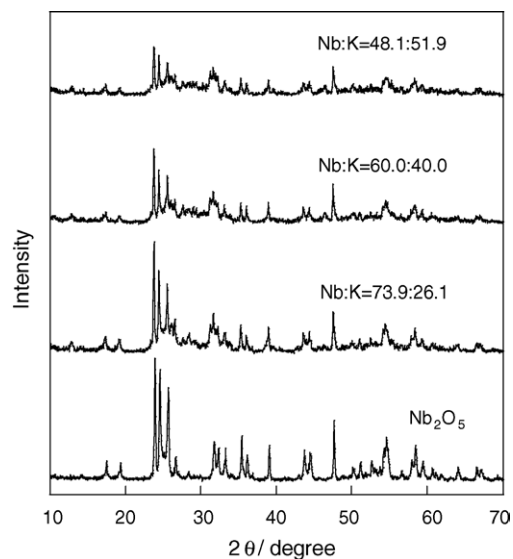


Fig. 2. The X-ray powder diffraction patterns of KNO₂ mixed Nb₂O₅.

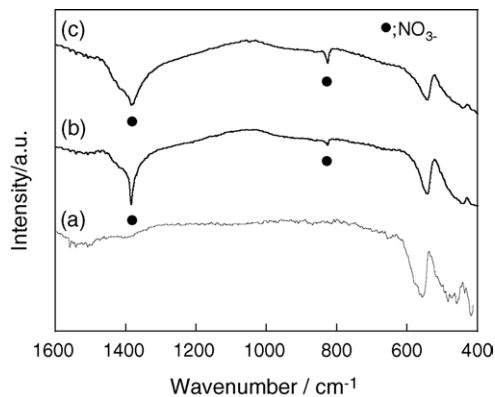


Fig. 3. FT-IR spectra of (a) Gd_2O_3 , (b) $0.595\text{Gd}_2\text{O}_3-0.405\text{KNO}_3$ and (c) $0.531\text{Gd}_2\text{O}_3-0.469\text{KNO}_3$.

contain NO_3^- at all. Although a peak shift in XRD and the IR bands corresponding to NO_3^- anion were not recognized at all, the K content in the KNO_2 mixed Nb_2O_5 system was the same as that of the mixing ratio from the X-ray fluorescence analysis. The results indicate the fact that KNO_3 produced from the oxidation of KNO_2 decomposes during preparation procedure.

The KNO_3 content dependencies of the electrical conductivity at 500°C for the Gd_2O_3 – KNO_3 series are presented in Fig. 4. The electrical conductivity increases with the KNO_3 content in the Gd_2O_3 – KNO_3 solids and shows the maximum conductivity at around $x = 0.405$. As the sample with $x = 0.469$ was the mixed phase (see Fig. 1), the highest conductivity of $10^{-1} \text{ S cm}^{-1}$ was obtained for the sample with $x = 0.405$ among the samples having the single solid solution phase. On the other hand, the conductivities for the KNO_2 mixed Nb_2O_5 series were independent of the mixing ratio of KNO_2 .

Fig. 5 shows the temperature dependencies of the conductivity for the $0.595\text{Gd}_2\text{O}_3-0.405\text{KNO}_3$ solid solution and the 0.4KNO_2 mixed $0.6\text{Nb}_2\text{O}_5$ solid, together with the data of $\text{K}^+-\beta''-\text{Al}_2\text{O}_3$ single crystal (in ab plane) and polycrystalline K_2SO_4 . The conductivity of the $0.595\text{Gd}_2\text{O}_3-0.405\text{KNO}_3$ solid solution is three orders of magnitude higher than that of KNO_2 mixed Nb_2O_5 , indicating the formation of the solid solution by introducing K^+ cation and NO_3^- anion simul-

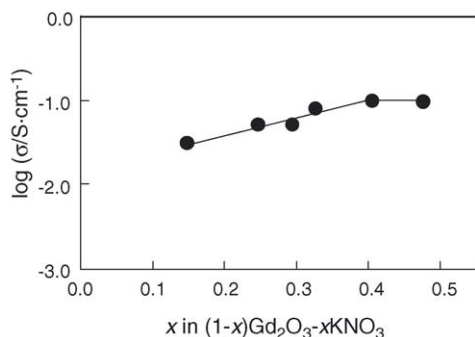


Fig. 4. The KNO_3 content dependence of the electrical conductivity for the Gd_2O_3 – KNO_3 series at 500°C .

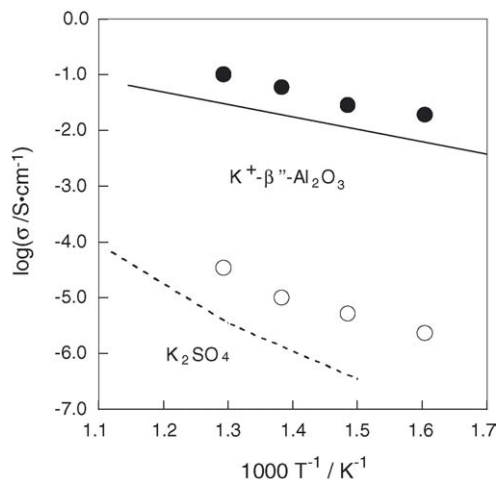


Fig. 5. The temperature dependencies of the conductivity for the $0.595\text{Gd}_2\text{O}_3-0.405\text{KNO}_3$ solid solution (●) and 0.4KNO_2 mixed $0.6\text{Nb}_2\text{O}_5$ (○) with the corresponding data of $\text{K}^+-\beta''-\text{Al}_2\text{O}_3$ single crystal [5] (in ab plane) (solid line) and polycrystalline K_2SO_4 [2] (dotted line).

taneously in the Gd_2O_3 crystal lattice. The conductivity of $0.595\text{Gd}_2\text{O}_3-0.405\text{KNO}_3$ is two times higher than that of the $\text{K}^+-\beta''-\text{Al}_2\text{O}_3$ single crystal (in ab plane) having the highest K^+ ion conductivity among the K^+ ion conducting solids reported so far. The solid solution formation is also effective to avoid decomposition of KNO_3 , which is clearly observed in the case for the Nb_2O_5 – KNO_3 series.

4. Conclusions

Two types of solid mixtures, KNO_2 mixed Gd_2O_3 and KNO_2 mixed Nb_2O_5 , were studied as the starting materials for the synthesis of new type of K^+ ion conducting solids. The formation of the solid solution with holding both K^+ cation and NO_3^- anion in the interstitial Gd_2O_3 crystal lattice was explicitly observed in the case for Gd_2O_3 – KNO_3 , and the K^+ ion conductivity was as high as $10^{-1} \text{ S cm}^{-1}$ at 500°C , while decomposition of the salt with K^+ occurred in the case for the KNO_2 mixed Nb_2O_5 series, resulting in the great conductivity decrease compared with the cases for the Gd_2O_3 – KNO_3 solids and the $\text{K}^+-\beta''-\text{Al}_2\text{O}_3$ single crystal (in ab plane). The results described above clearly support the idea that the Gd_2O_3 – KNO_3 solid is the superior K^+ ion conducting solid solution electrolyte which contains not only K^+ cation but also NO_3^- anion in the polycrystalline C-type cubic Gd_2O_3 crystal lattice.

Acknowledgment

This work was supported by the Industrial Technology Research Grant Program in 2002 (Project ID: 02A27004c) from the New Energy and Industrial Technology Development Organization (NEDO) based on funds provided

by the Ministry of Economy, Trade and Industry, Japan (METI).

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